



## Review

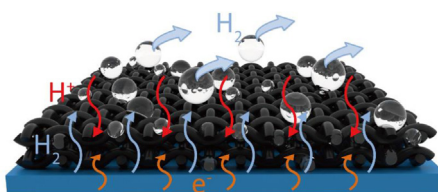
## Recent progress in electrode fabrication for electrocatalytic hydrogen evolution reaction: A mini review

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## HIGHLIGHTS

- HER performance depends on electron transfer, proton diffusion and bubble release.
- A wide range of support materials.
- Critical substrate parameters include porosity, electrical conductivity, and chemical stability.
- Electrode fabrication by 3D printing.
- Superaerophobic surface.

## GRAPHICAL ABSTRACT



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## ABSTRACT

Electrochemical hydrogen evolution reaction (HER) is of great importance for high-efficiency, low-cost production of hydrogen. As the HER activity of catalysts has been significantly improved in recent years, the routings from catalysts to electrodes represent a key step for practical HER applications. In an HER electrode, hydrogen reduction is accomplished by electron transfer, proton diffusion and bubble release. These processes should be taken into account in the design, engineering and fabrication of the electrode. In this review, we begin with a discussion of the correlation between these important issues and the overall electrochemical HER performance, and then summarize recent progress in the design and fabrication of HER electrodes, with a focus on the effects of electrode structure, electrolyte penetration, ion diffusion and bubble adhesion/release on the HER performance. We conclude with a perspective of strategies for further enhancement and the critical challenges for HER electrode fabrication.

## 1. Introduction

Development of renewable energy conversion and storage technologies has been hailed as a viable strategy to reduce our dependence on fossil fuels, such as coal and gasoline, and the emissions of waste gases such as carbon dioxide, sulfur dioxide and nitrogen oxides [1–3]. Hydrogen, as a clean energy carrier with a high energy density, has been generally recognized as a promising alternative to fossil fuels [4–7]. To this end, hydrogen evolution reaction (HER) in water electrolysis has been attracting extensive attention, where high-purity hydrogen is produced under ambient conditions. HER primarily involves two

routes, the Volmer–Tafel pathway, and the Volmer–Heyrovsky pathway [8]. The HER efficiency is significantly dependent on the selection of catalysts. Currently, platinum and ruthenium oxide-based materials are the catalysts of choice for HER in both acid and base media [9–11], but high costs, limited availability and natural scarcity have hindered their wide-spread applications.

Thus, significant efforts have been devoted to the development of high-efficiency HER catalysts based on earth-abundant, cost-effective materials. Among these, transition metal sulfides, phosphides, oxides, carbides and nitrides have been found to demonstrate high catalytic activities towards HER [12–17]. For example, Tian et al. [18] reported

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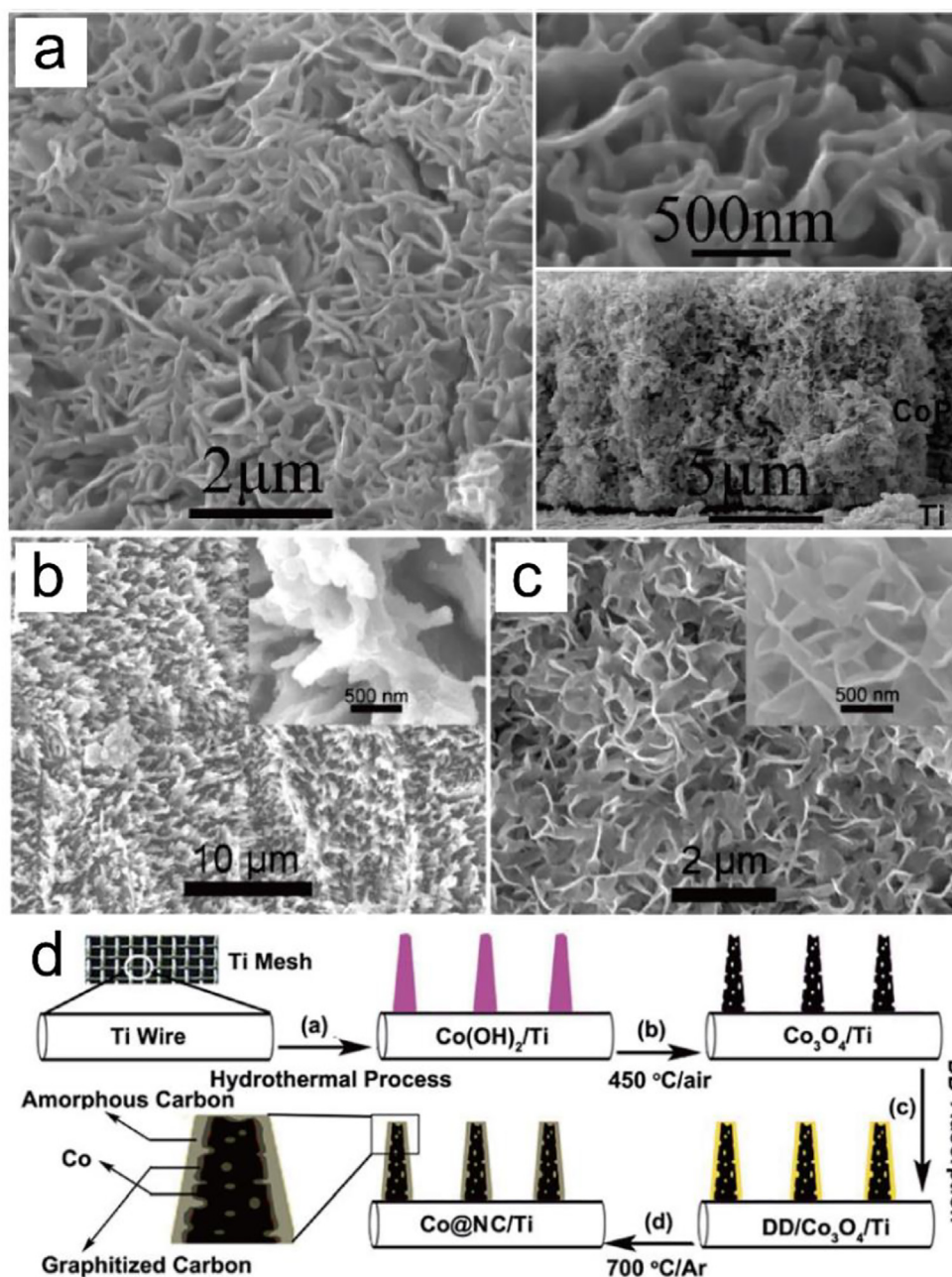
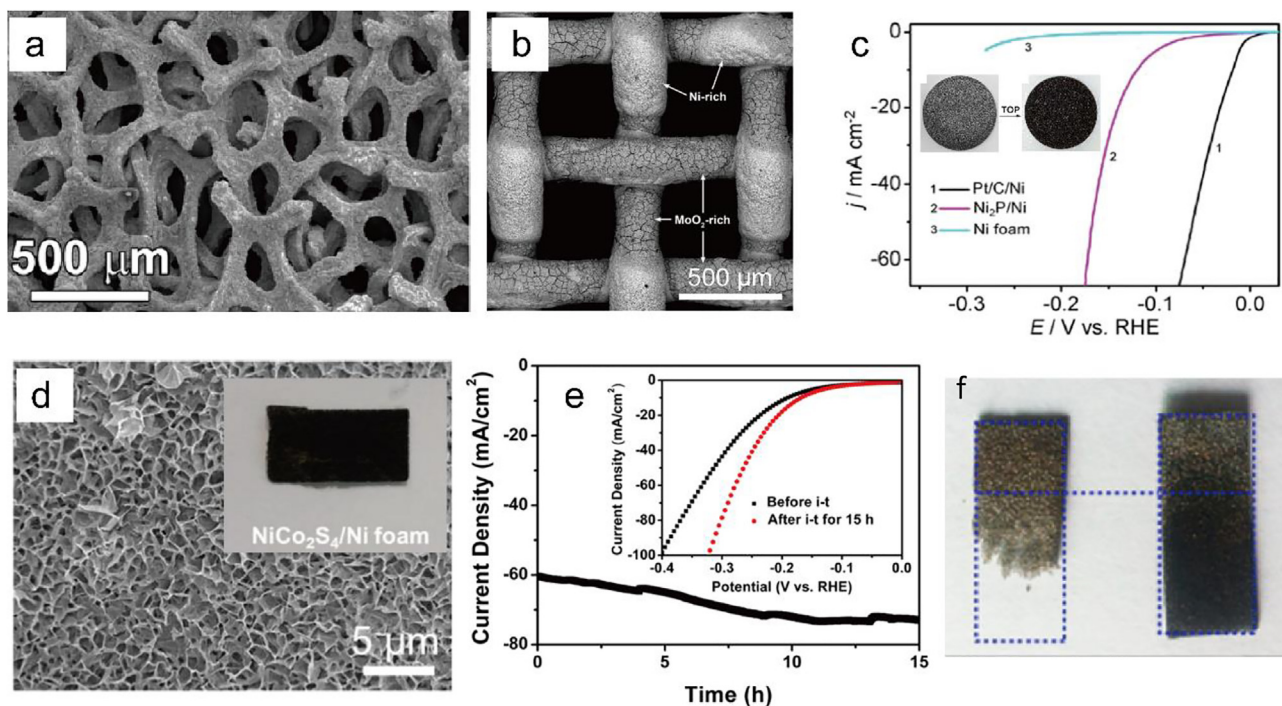


Fig. 1. SEM images of (a) CoP/Ti electrode, (b) Ti plate and (c) MoS<sub>2</sub> NAs/Ti electrode. Insets to panels (b, c) are the corresponding high-resolution images. Panel (a) reproduced with permission from Ref. [42] © American Chemical Society. Panels (b, c) reproduced with permission from Ref. [44] © Elsevier. (d) Schematic of the synthesis of Co@NC/Ti. Panels (d) reproduced with permission from Ref. [40] © Elsevier.

that self-supported nanoporous cobalt phosphide nanowire arrays can serve as highly efficient HER catalysts, which exhibited a low overpotential ( $\eta_{10}$ ) of  $-67$  mV to reach the current density of  $10 \text{ mA cm}^{-2}$  (close to that of Pt). Similarly, Liang et al. [19] prepared iron phosphide nanorod arrays as cost-effective HER catalysts, with a lower  $\eta_{10}$  of  $-58$  mV in  $0.5 \text{ M H}_2\text{SO}_4$ . As the intrinsic activity of such catalysts has been significantly improved in the past decades, current efforts have been devoted to implementing these catalysts for HER applications through electrode design and fabrication.

Traditional HER electrodes are fabricated by coating a layer of active materials on the electrode surface, which should be chemically stable and electrically conductive. The active materials and the substrate serve as a porous layer for HER catalysis and as a current collector, respectively [20]. In the electrode, there are three major processes during HER, electron transfer, proton diffusion and bubble

release, that largely dictate the HER performance. Therefore, the fabrication of an effective HER electrode should satisfy three principles. First, the catalysts used should possess a high intrinsic electrocatalytic activity, which can facilitate electron transfer between protons and electrode. Second, the electrode (and catalyst) should have a highly porous structure, which can provide sufficient channels for proton diffusion and bubble transport inside the electrode. Third, the electrode surface should promote the release of hydrogen bubbles, since bubbles can block ion diffusion and reduce the effective surface area of the electrode. With an abundant pore structure and high electrical conductivity, nickel foam, carbon cloth, graphene film, graphene aerogel, carbon aerogel, and carbon felt have been used rather extensively as electrode substrates [21–26]. For instance, Shi et al. prepared an HER electrode based on MoSe<sub>2</sub> supported on graphene-carbon nanotube aerogel (GCA-MoSe<sub>2</sub>), and found that GCA significantly reduced the



**Fig. 2.** (a) SEM image of a nickel foam. Panel (a) reproduced with permission from Ref. [48] © American Chemical Society. (b) SEM image of a nickel mesh. Panel (b) reproduced with permission from Ref. [49] © Elsevier. (c) Polarization curves of Ni<sub>2</sub>P/Ni, Pt/C/Ni, and bare Ni foam at the potential scan rate of 5 mV s<sup>-1</sup> in 0.5 M H<sub>2</sub>SO<sub>4</sub>. Insets are the photographs of the bare Ni foam before (left) and after (right) being treated with TOP at elevated temperature. Panel (c) reproduced with permission from Ref. [56] © American Chemical Society. (d) SEM image of the NiCo-LDH/Ni foam. Inset is the corresponding photograph of the NiCo-LDH/Ni foam. Panel (d) reproduced with permission from Ref. [59] © Elsevier. (e) Current-time plots of Ni foam electrode at the applied potential of -0.43 V (vs RHE) in 0.5 M H<sub>2</sub>SO<sub>4</sub> for up to 15 h. Inset are the corresponding polarization curves before and after 15 h's continuous operation. (f) Photographs of the Ni foam (left) dipped in 0.5 M H<sub>2</sub>SO<sub>4</sub> solution for 15 h and (right) after i-t test at -0.43 V vs RHE in 0.5 M H<sub>2</sub>SO<sub>4</sub> for 15 h. Panels (e, f) reproduced with permission from Ref. [60] © American Chemical Society.

onset potential ( $E_{\text{onset}}$ ) from -198 to -113 mV vs. RHE, and  $\eta_{10}$  from -297 to -228 mV in 0.5 M H<sub>2</sub>SO<sub>4</sub>, as ion/electrolyte transport at electrode interface was facilitated [27].

Notably, high-efficiency catalysts alone are not sufficient in the fabrication of an HER electrode, as the contact between the electrode and electrolyte can be hindered by the adhesion of hydrogen bubbles on the electrode surface, resulting in an increase of ohmic drop and a decrease of HER efficiency [28,29]. For example, in the evaluation of the HER performance of a cathode at the constant potential of -0.3 V vs. RHE, Hou et al. [30] observed a fluctuation of the HER current that was closely related with the accumulation and release of H<sub>2</sub> bubbles on the electrode surface. Similarly, Perera et al. [31] investigated the dynamic behaviors of hydrogen bubbles and found that the transient current response was dominated by the bubble nucleation, growth, and release. To minimize the detrimental effect induced by H<sub>2</sub> bubbles, the concept of “superaerophobic” surface has been introduced for HER electrode design. Several studies have shown that a “superaerophobic” surface can be constructed by changing the three-phase contact line (TPCL), which is crucial for the interactions between H<sub>2</sub> bubbles and solid surfaces [28,29,32,33]. In this design, the TPCL can be adjusted by controlling the nanostructures on the electrode surface during electrode fabrication, which effectively diminishes the adhesion force of bubbles and improves the HER performance. Although a number of remarkable catalysts and electrodes with “superaerophobic” surfaces have been developed, the evolution from catalysts to electrodes needs further exploration.

Prior reviews have mainly focused on the studies of the catalyst materials, analysis of the catalytic mechanism and electrocatalytic activity [34–38]. Yet, reports of the fabrication of HER electrodes have been scarce. In this review we will first discuss the relationship between electrode structure, bubble adhesion/release, electrolyte penetration,

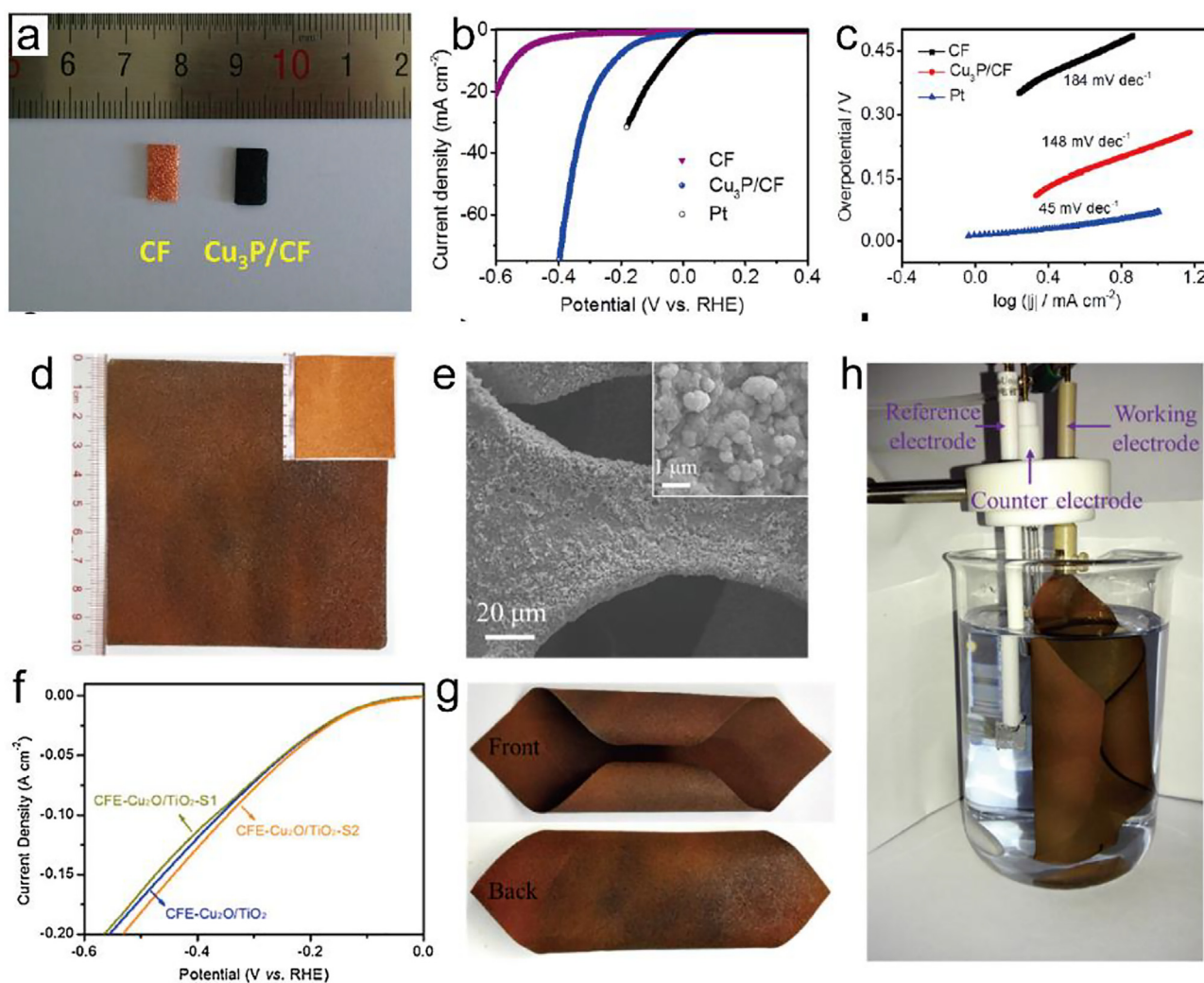
ion diffusion and HER performance of the electrode, summarize recent progress in the design and fabrication of HER electrodes, describe approaches to further improve the electrode performance, and finally include a prospective of the critical challenges for HER electrode design.

## 2. Transition metal-based electrodes

### 2.1. Titanium plate/mesh-based electrodes

Because of excellent electrical conductivity and high corrosion resistance to acid and alkaline media, titanium plates have been chosen as electrode substrates and current collectors in recent studies [39–41]. To prepare HER electrodes, a layer of catalysts is usually coated on the surface of a Ti plate, such as CoP nanosheet arrays, Ni<sub>2</sub>P nanoparticles, FeP nanoparticles/nanorods, CoSe<sub>2</sub> nanosheets, and MoS<sub>2</sub> nanosheets [39,42–46]. For example, Pu et al. [42] grew a CoP nanosheet array onto a Ti plate by electrodeposition at room temperature, which was then subject to low-temperature phosphidation. The as-prepared HER electrode exhibited an  $E_{\text{onset}}$  of -40 mV vs. RHE and  $\eta_{10}$  of -90 mV in 0.5 M H<sub>2</sub>SO<sub>4</sub> (Fig. 1a). Similarly, Shi et al. [44] prepared an HER electrode by depositing a layer of MoS<sub>2</sub> nanosheet arrays onto a Ti plate (MoS<sub>2</sub> NAs/Ti) through a simple hydrothermal and calcination process, which displayed a small  $E_{\text{onset}}$  of -90 mV, and  $\eta_{10}$  of -108 mV in 0.5 M H<sub>2</sub>SO<sub>4</sub> (Fig. 1b, c). The MoS<sub>2</sub> NAs/Ti electrode also exhibited high stability with a negligible loss of current density during continuous cyclic voltammetric and chronoamperometric tests.

Although a high catalytic activity is generally observed by coating a layer of nanostructured catalysts on the metal plate surface, it should be pointed out that the plate electrode is not conducive to mass transport (e.g., ion diffusion and bubble release). According to the principles of



**Fig. 3.** (a) Photograph of blank CF (left) and  $\text{Cu}_3\text{P}/\text{CF}$  (right), (b) polarization curves and (c) the corresponding Tafel plots of  $\text{Cu}_3\text{P}/\text{CF}$ , CF, and commercial Pt. Panels (a–c) reproduced with permission from Ref. [61] © American Chemical Society. (d) Digital photograph and (e) SEM image of the as-prepared  $\text{Cu}_3\text{P}/\text{CF}$  electrode (the corresponding insets are of the untreated CF). (f) Polarization curves of the  $\text{Cu}_3\text{P}/\text{CF}$  electrode, (g) digital photograph of the  $\text{Cu}_3\text{P}/\text{CF}$  electrode bended into a specific shape, and (h) digital photograph of a HER set up in 1 M KOH using the  $\text{Cu}_3\text{P}/\text{CF}$  as the working electrode, Pt as the counter electrode and Ag/AgCl as the reference electrode. Panels (d–h) reproduced with permission from Ref. [62] © Elsevier.

electrode design, proton supply and bubble release can dominate the HER process. In a computational study based on finite elements simulation methods, Carneiro-Neto et al. [47] investigated the pH change at the electrode–electrolyte interface during HER, which was closely related to the current density. They found that the interfacial pH was 2–3 units higher than that in the bulk solution. This indicates that the  $\text{H}^+$  concentration was 2–3 orders of magnitude lower near the electrode surface, corresponding to a 118–177 mV overpotential according to the Nernst equation. Therefore, to alleviate the limitation of ion diffusion and bubble releasing, titanium meshes have been used instead as alternative substrates for HER electrodes due to its porous structure. For example, in a recent study [40], N-doped carbon-encapsulated cobalt nanorod arrays were grown on a titanium mesh ( $\text{Co}@ \text{NC}/\text{Ti}$ ) as a HER electrode by a two-step procedure involving hydrothermal synthesis of  $\text{Co}_3\text{O}_4$  nanorods followed by thermal reduction to metallic cobalt (Fig. 1d). The obtained  $\text{Co}@ \text{NC}/\text{Ti}$  electrode exhibited an  $E_{\text{onset}}$  of  $-56$  mV, which was close to that of commercial Pt/C ( $-12$  mV); and the current densities increased sharply by 10 folds from 10 to  $100 \text{ mA cm}^{-2}$  when the  $\text{Co}@ \text{NC}/\text{Ti}$  electrode potential was swept by less than 80 mV from  $-106$  mV to  $-184$  mV.

## 2.2. Nickel foam/mesh-based electrodes

Nickel foam/mesh has also been used extensively as an electrode substrate for HER electrodes (Fig. 2a and b), due to its excellent electrical conductivity, three-dimensional cross-linked network and hierarchical porous structure [48–50]. In addition, active materials in the forms of nanowires, nanoparticles, nanoflakes and nanorods can be in situ synthesized on the surface of nickel foam/mesh, such as nickel phosphides ( $\text{Ni}_2\text{P}$ ) [51], nickel sulfide ( $\text{Ni}_3\text{S}_2$ ) [52] and nickel alloys (such as Ni–B, Ni–Mo) [53,54] by chemical vapor deposition, hydrothermal treatment, low temperature phosphidation, and electrodeposition, which minimizes deactivation resulting from the reversible formation of nickel hydride species under alkaline condition [55]. Notably, a polymer binder is not needed when catalysts are directly in situ grown on the skeleton of nickel foam/mesh, minimizing the overpotential loss that arises from the block of active sites, the inhibition of ion diffusion, and the electrocatalyst/electrode interfacial resistance [51]. For instance, Shi et al. [56] prepared a  $\text{Ni}_2\text{P}$  nanosheets/Ni foam composite electrode through a facile chemical conversion pathway by using surface-oxidized Ni foam as a precursor and low concentration of trioctylphosphine as a phosphorus source, and the obtained  $\text{Ni}_2\text{P}/\text{Ni}$  electrode exhibited an improved HER performance with an  $E_{\text{onset}}$  of ca.

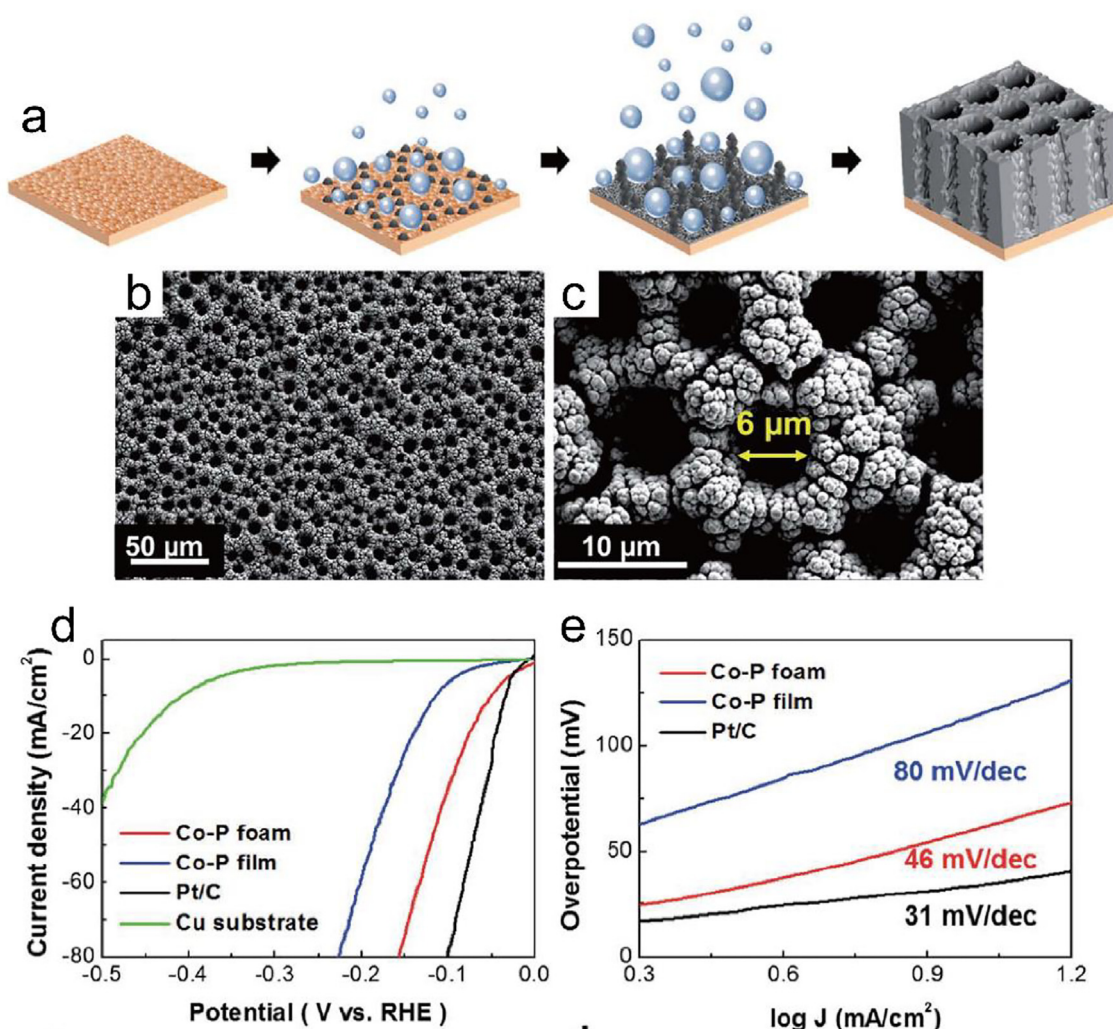


Fig. 4. (a) Schematic illustration of the synthesis of the Co-P foam, (b, c) SEM images of the Co-P foam at varied magnifications, (d) polarization and (e) Tafel curves of Co-P foam, Co-P film, Pt/C and Cu substrate. Reproduced with permission from Ref. [64] © Royal Society of Chemistry.

–80 mV in acid, significantly better than that of bare Ni foam (Fig. 2c).

In addition to nickel phosphides and nickel alloys, other transition metal sulfides and phosphides, such as CoP [57], FeP [20], MoS<sub>2</sub> [58], have also been used as active materials anchored on the surface of nickel foam substrates to improve the HER activity and durability of the electrode. For example, Ma et al. [59] prepared a NiCo<sub>2</sub>S<sub>4</sub>/Ni foam electrode by growing a layer of ultrathin NiCo<sub>2</sub>S<sub>4</sub> nanoflakes onto the surface (Fig. 2d), and the electrode exhibited a low  $E_{\text{onset}}$  of –17 mV and long cycling stability (100,000 s) in alkaline solution. Nevertheless, the sustainability of nickel-based electrodes in acidic solutions remains challenging, because of easy dissolution of nickel by acid. To mitigate this issue, Lu et al. [60] described an effective method to stabilize the electrode by applying an overpotential as compared to the equilibrium potential of Ni<sup>0</sup>/Ni<sup>2+</sup> to the HER electrode, and the resulting nickel foam electrode exhibited an excellent and stable HER activity with an  $E_{\text{onset}}$  of –84 mV,  $\eta_{10}$  of –210 mV, and prominent electrochemical durability (> 5 d) in acidic electrolyte (Fig. 2e, f).

### 2.3. Copper foam/mesh-based electrodes

In comparison to nickel foam/mesh, copper foam/mesh has a lower cost and higher electrical conductivity, and thus is a highly favorable material as a HER electrode substrate. However, due to the low electrocatalytic activity of the copper foam substrate, a layer of catalytically active materials, such as Cu/Cu<sub>2</sub>O nanowires, NiCoP@Cu<sub>3</sub>P forests,

CuN nanotubes and Cu<sub>3</sub>P nanoarrays, has been anchored on the surface of copper foam through chemical or electrochemical methods, e.g., chemical oxidation, calcination and sequential phosphidation, and electrodeposition [61,63]. For example, Yu et al. [63] prepared Cu nanowires coated with FeNi layered double hydroxide nanosheets on a copper foam, which involved a chemical oxidation process to produce Cu(OH)<sub>2</sub> nanowires on the copper foam, transformation to CuO nanowires by calcination, and electrochemical deposition of FeNi layered double hydroxide nanosheets onto the nanowires. To simplify the preparation procedure, Hou et al. [61] fabricated an HER electrode by directly phosphatizing the copper foam at 300 °C using NaH<sub>2</sub>PO<sub>2</sub> to form a layer of semimetallic Cu<sub>3</sub>P nanoarrays. However, despite an apparent improvement of the HER performance, the as-prepared electrode exhibited a relatively high  $\eta_{10}$  of –222 mV in alkaline media, much larger than that of commercial Pt/C (–57 mV) (Fig. 3a–c). Interestingly, Long et al. [62] prepared a copper foam electrode decorated with a Cu<sub>2</sub>O/TiO<sub>2</sub> composite through a facile etch and heat treatment, and the obtained electrode achieved a low  $\eta_{10}$  of –114 mV in alkaline media without iR correction (only slightly weaker than that of a Pt mesh ( $\eta_{10}$  = –82 mV)). Scale-up experiments further demonstrated the feasibility in large-scale preparation of this electrode (Fig. 3d–h). More importantly, the as-prepared electrode even delivered an improved HER performance after the stability tests, probably due to the formation of Cu<sub>2</sub>O/TiO<sub>2</sub> heterojunctions which facilitated electron transfer from Cu<sub>2</sub>O to TiO<sub>2</sub> and hence led to enhanced HER on TiO<sub>2</sub>.

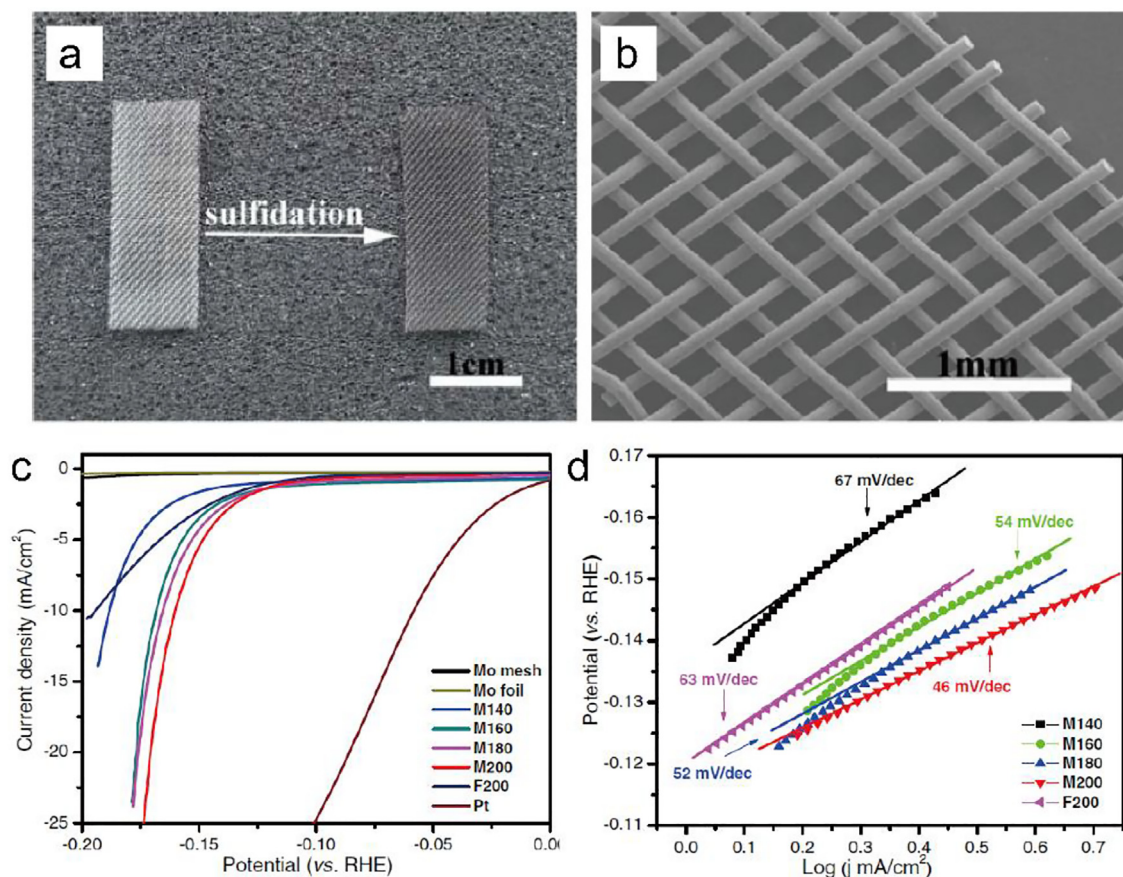


Fig. 5. (a) Photograph of a Mo mesh before (left) and after (right) hydrothermal treatment in aqueous thiourea solution, (b) SEM image of the Mo mesh, (c) polarization curves and (d) the corresponding Tafel plots for the Mo mesh-based electrodes. Reproduced with permission from Ref. [65] © Elsevier.

In addition to chemical oxidation, calcination and low temperature phosphidation, electrodeposition at a high current density is also a facile approach to the preparation of porous HER electrodes. For example, Oh et al. [64] prepared forest-like NiCoP@Cu<sub>3</sub>P supported on a copper foam for HER by electrodeposition at the applied cathodic current density of 3 A cm<sup>-2</sup>. During the electrodeposition process, the high current density induced fast deposition of Co and P on the substrate, and the rapidly generated hydrogen bubbles acted as a dynamic template for the skeleton Co-P structure by inhibiting the deposition of Co-P on the substrate surface, eventually leading to the formation of a highly porous Co-P foam structure (Fig. 4a-c). The as-prepared electrode exhibited a low  $\eta_{10}$  of -50 mV in 0.5 M H<sub>2</sub>SO<sub>4</sub> and -131 mV in 1 M KOH, comparable to those of commercial Pt/C (-33 mV in 0.5 M H<sub>2</sub>SO<sub>4</sub>, and -80 mV in 1 M KOH). In addition, the electrode also showed high stability with a small increase of  $\eta_{10}$  from -50 to -63 mV after 2000 cycles.

Taken together, these studies demonstrated the feasibility of the fabrication of highly efficient HER electrode based on copper foam. However, it should be noticed that the corrosion resistance of copper foam/mesh is generally lower than that of nickel metal; thus the sustainability of copper foam/mesh as HER electrodes needs to be further improved.

#### 2.4. Other metal-based electrodes

In addition to the materials discussed above, other metals such as stainless steel, gold and silver, have also been used as HER electrode substrates [66-68]. Considering the inert property of silver and gold, active catalysts are generally deposited onto the substrate surfaces. For example, Uosaki et al. [68] prepared an HER electrode by depositing

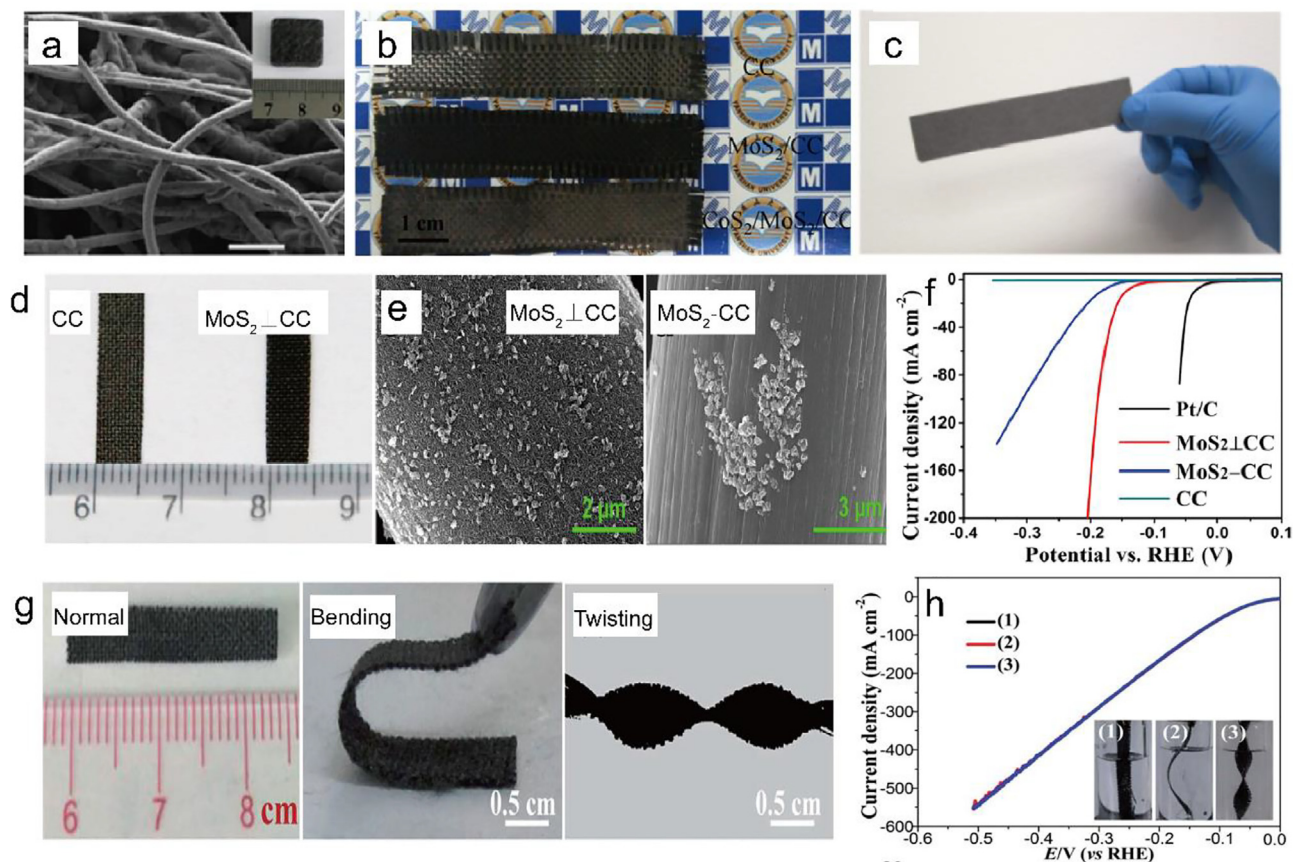
boron nitride nanosheets on a gold substrate, and the electrode exhibited an overpotential only -30 mV greater than Pt/C to reach the current density of 5 mA cm<sup>-2</sup> in 0.5 M H<sub>2</sub>SO<sub>4</sub>. Kiani et al. [69] fabricated a platinum-coated nanoporous gold film as HER electrode, which delivered a current density of 198 mA cm<sup>-2</sup> at -0.8 V vs. Ag/AgCl and a mass activity of 1940 mA  $\mu$ g<sup>-1</sup>. However, the high costs and low reserves of precious metals (silver, gold and platinum) have severely limited their practical applications. Several studies have shown that earth-abundant, low-cost metal substrates can be used for the preparation of HER electrodes, such as stainless steel mesh, stainless steel rod, molybdenum mesh [65-67]. For example, Xu et al. synthesized three-dimensional (3D) arrays of MoS<sub>2</sub> nanosheets on a Mo mesh as an HER electrode by hydrothermal treatment in an aqueous thiourea solution, which exhibited an  $\eta_{10}$  of -160 mV in acid (Fig. 5) [65]. In spite of a low cost and apparent HER performance, the properties such as corrosion resistance and HER kinetics have rarely been studied, and their feasibility and sustainability in HER remain to be explored.

### 3. Porous carbon-based electrodes

#### 3.1. Carbon cloth/paper/felt-based electrodes

Compared to metal-based electrodes, carbon materials have the unique advantages of low cost, high specific surface area, abundant pore structure, tunable elemental doping, and good chemical stability, making them promising electrode supports. In fact, the HER performance of carbon-based electrodes can be effectively improved through deliberate structural engineering.

Among the different carbon substrates, carbon cloth, carbon paper and carbon felt are the most commonly used electrode supports with



**Fig. 6.** (a) SEM of carbon felt. Inset is a corresponding photograph. Panel (a) reproduced with permission from Ref. [70] © American Chemical Society. (b) Photograph of carbon cloth. Panel (b) reproduced with permission from Ref. [71] © Elsevier. (c) Photograph of carbon paper. Panel (c) reproduced with permission from Ref. [72] © American Chemical Society. (d) photograph of blank CC and MoS<sub>2</sub>-CC, (e) SEM images of MoS<sub>2</sub>-CC and MoS<sub>2</sub>-CC, and (f) HER polarization curves of varied electrodes. Panels (d-f) reproduced with permission from Ref. [73] © American Chemical Society, (g) Photographs and (h) polarization curves of carbon cloth under the various distorted states. Panels (g, h) reproduced with permission from Ref. [74] © Royal Society of Chemistry.

high electrical conductivity, interconnected pores and chemical stability. These features endow the HER electrode with high durability, facile ion diffusion and increased catalytic activity (Fig. 6a–c). In the traditional electrode fabrication, the nanocatalyst power is dispersed into a solution with the addition of a binder to prepare the catalyst ink, which is then coated onto the carbon substrates. For example, Chanda et al. [75] prepared an HER electrode by coating NiFe<sub>2</sub>O<sub>4</sub> catalysts onto a carbon substrate using poly(phenylene oxide) ionomer, polytetrafluoroethylene and Nafion®, and observed that the HER electrode with the poly(phenylene oxide) ionomer achieved the best performance. However, the use of an electrical insulating binder may decrease electrical conductivity of the electrode, block catalytic active sites and hinder ion diffusion within electrodes, hence resulting in a reduced electrocatalytic performance.

In addition, aggregation and peeling off of catalysts from substrates may occur [76]. Recently, many studies have shown that binder-free HER electrodes can be prepared by direct growth of catalysts on the carbon cloth/carbon paper/felt substrates using hydrothermal, solvothermal or electrochemical methods. For example, Zhang et al. [73] prepared a binder-free electrode by growing MoS<sub>2</sub> nanosheets vertically on carbon cloth (MoS<sub>2</sub>-CC), and a binder-based electrode by physically coating MoS<sub>2</sub> on carbon cloth using Nafion (MoS<sub>2</sub>-CC) (Fig. 6d–e), and found that MoS<sub>2</sub>-CC delivered a current density of 200 mA cm<sup>-2</sup> at the overpotential of -205 mV in acidic media, whereas it was only ca. 20 mA cm<sup>-2</sup> for MoS<sub>2</sub>-CC (Fig. 6f). This can be attributed to the well-dispersed MoS<sub>2</sub> on the porous skeleton that facilitated ion diffusion in the binder-free electrode [77]. In another study, Wang et al. [74] reported a new type of Ni<sub>2</sub>P-CoP hybrid nanosheet array on carbon cloth

(Ni<sub>2</sub>P-CoP HNSAs/CC), which exhibited a high HER performance with a small  $\eta_{10}$  of -85 mV and high stability with only a 12.49% loss of current density after 14 h's discharging at a constant potential of -0.14 V vs. RHE. Importantly, the as-developed cathode also showed an excellent flexibility and virtually no change of the HER performance under various bending and twisting states. These results confirmed the feasibility and reliability of this electrode in practical applications (Fig. 6g and h).

It should be noted that although many studies have demonstrated satisfactory performance of binder-free electrodes in all-pH media, the HER catalysts are usually adhered on the surfaces of porous carbon or carbon felt substrate, and the interconnected pore structures are not sufficiently utilized. This may compromise the eventual performance.

### 3.2. Graphene/Carbon nanotube film-based electrodes

Considering the low utilization rate of porous carbon cloth/paper/felt, the electrodes can be constructed by mixing HER catalysts, such as WS<sub>2</sub>, and Co nanoparticles, with graphene or carbon nanotubes, and then filtering the mixture to obtain a thin film in which the catalysts are homogeneously dispersed [76,78]. For example, Hou et al. [78] prepared a Co nanoparticles@nitrogen-doped graphene film (Co@NGF) as a flexible, porous electrode. The procedure involves three steps: (a) mixing a graphene oxide suspension with a dicyandiamide-cobalt complex to form a homogenous mixture; (b) filtration of the mixture to obtain a nanostructured film; and (c) annealing of the prepared film at 700 °C in an Ar atmosphere to obtain Co@NGF. The resulting film electrode achieved an  $E_{\text{onset}}$  of -14 mV and  $\eta_{10}$  of -124.6 mV in acidic

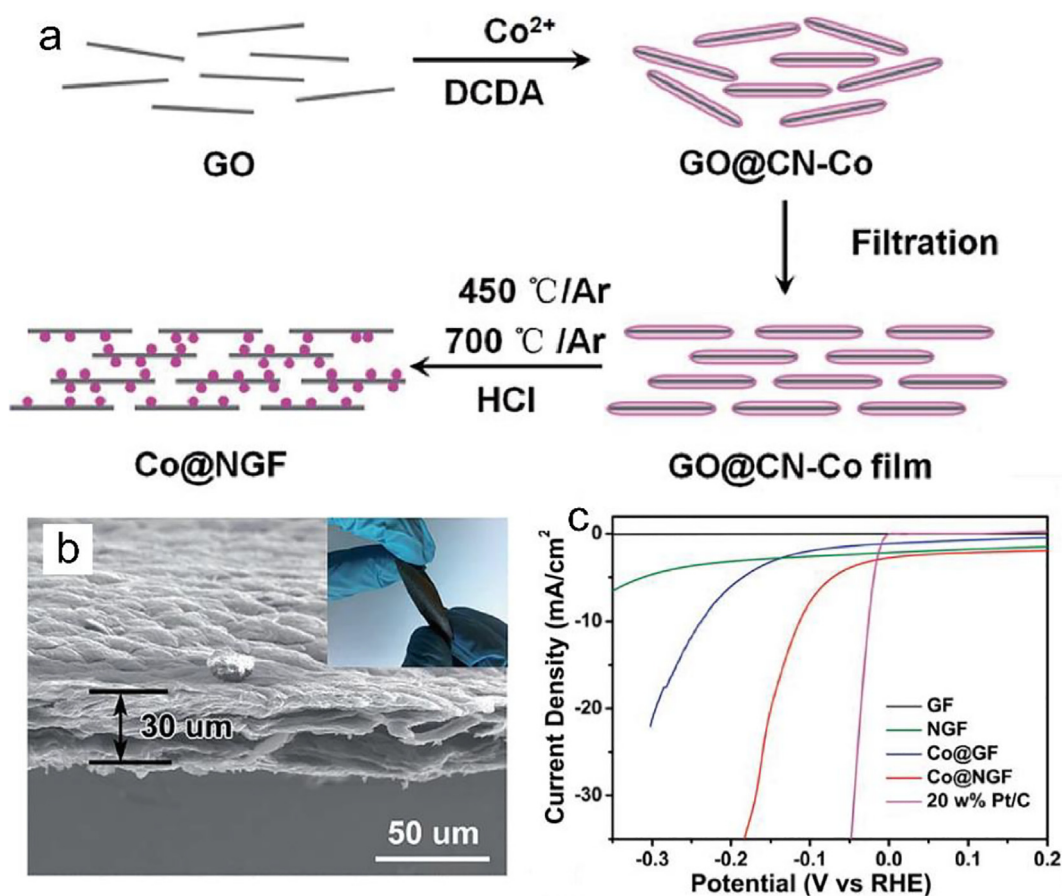


Fig. 7. (a) Schematic illustration of the synthesis and (b) SEM image of Co@NGF, (c) HER polarization curves of varied electrodes. Inset to (b) is a photograph of the Co@NGF electrode. Reproduced with permission from Ref. [78] © Royal Society of Chemistry.

media (Fig. 7). Apparently, the catalyst electrodes based on graphene and carbon nanotube have a sufficient catalyst/electrode contact area and high catalyst utilization rate due to the homogeneously dispersed catalysts. In addition, in combination with the high specific surface area and high porosity, the catalyst electrode can obtain an extended exposure of active sites and facilitated ion diffusion within the electrode.

### 3.3. Graphene/carbon nanotube/carbon fiber aerogel-based electrodes

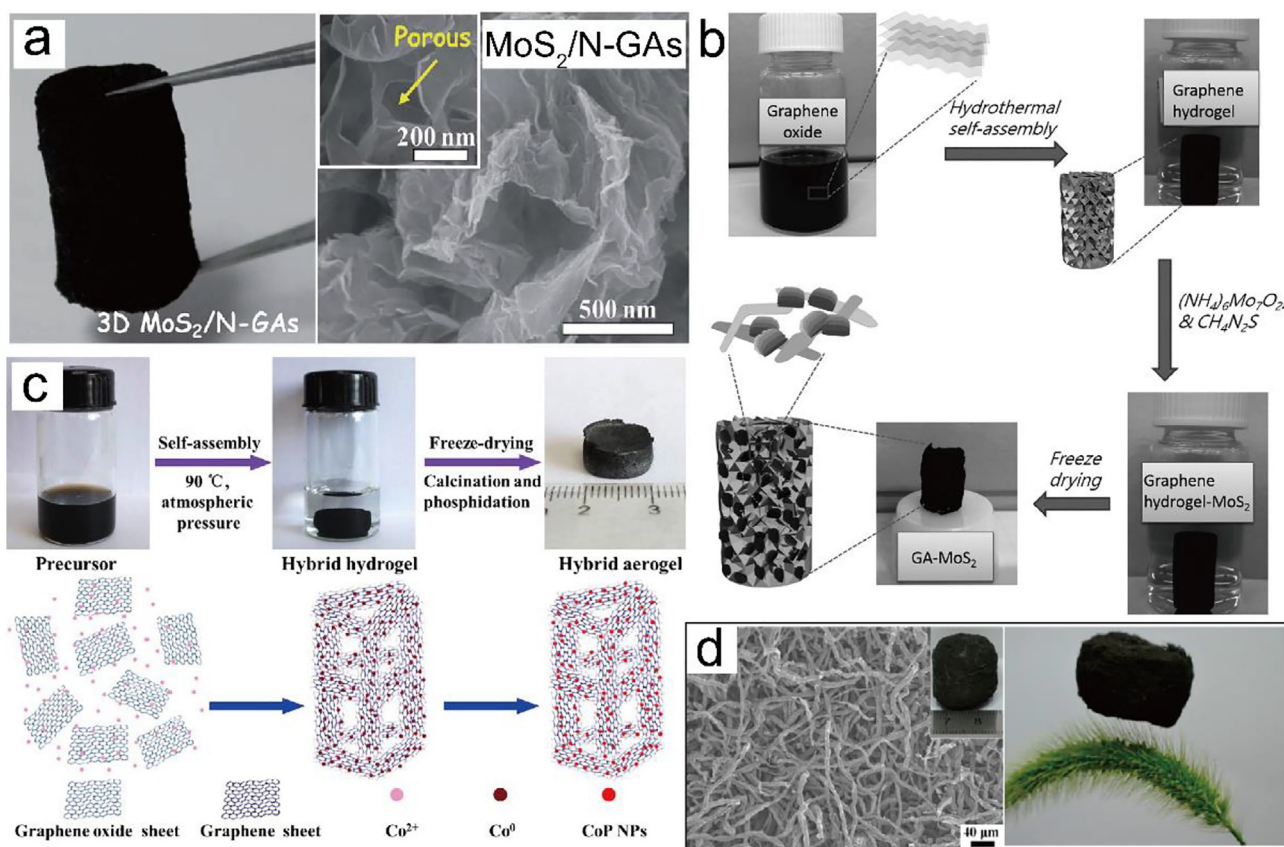
3D carbon materials such as graphene aerogel, carbon nanotube aerogel and carbon fiber aerogel can provide a high specific surface area (and electroactive surface area), abundant 3D interconnected network, enhanced accessibility by electrolytes, and multidimensional electron/ion transport channels (Fig. 8a), as compared to carbon cloth/carbon paper and graphene/carbon nanotube films [79,81]. These features make them promising supports in HER electrode fabrication. Typically, HER electrodes based on graphene aerogels are prepared in two steps: (1) self-assembly of graphene/carbon nanotube/carbon fiber hydrogel/aerogel by direct pyrolysis or hydrothermal method, and (2) decoration of catalysts on the obtained carbon support. For example, Zhao et al. [79] prepared 3D graphene aerogel-supported MoS<sub>2</sub> nanosheets (GA-MoS<sub>2</sub>) as an HER electrode by (a) hydrothermal self-assembly to form a graphene hydrogel, (b) a second hydrothermal treatment of the hydrogel in an aqueous solution containing ammonium heptamolybdate and thiourea to grow the MoS<sub>2</sub> nanosheets onto the hydrogel surface, and (c) freeze-drying to obtain the final product (Fig. 8b). To simplify the preparation procedure, Zhang et al. [21] fabricated a cobalt phosphide nanoparticles-decorated 3D graphene aerogel electrode. Experimentally, graphene oxide and Co<sup>2+</sup> were self-assembled to form a graphene hydrogel, which was then freeze-dried

and thermally treated and phosphidated to obtain cobalt phosphide nanoparticles-decorated 3D graphene aerogel (Fig. 8c). Zhang et al. [80] synthesized a cotton wool-derived carbon fiber aerogel as an electrode support by direct carbonization, instead of hydrothermal, solvothermal and freeze dry, significantly simplifying the preparation procedure of aerogels (Fig. 8d). Considering the high exposure of active sites, and multidimensional transport channels for electron and ions, these aerogel electrodes exhibited an improved HER performance. For example, Xu et al. [82] synthesized penroseite (Ni, Co)Se<sub>2</sub> nanocages anchored on a 3D graphene aerogel as HER electrodes, which exhibited an  $\eta_{10}$  of  $-156$  mV in 1 M KOH, only 28 mV higher than that of Pt/C. Nevertheless, the pore structures of aerogel electrodes may collapse during actual, long-term operation, especially for the electrodes prepared by hydrothermal and freeze drying processes, due to the limited  $\pi$ - $\pi$  interaction between carbon precursors.

### 3.4. Carbon foam-based electrodes

Different from the procedure for the preparation of graphene/carbon nanotube aerogels, one commonly used method to prepare carbon foam is direct pyrolysis of porous precursors, such as citric acid monohydrate, melamine, glucose and polyimide [83–86]. During the pyrolysis process, the precursors are dehydrated and polymerized, and the rapid release of H<sub>2</sub>O molecules leads to the formation of 3D carbon foam networks [84]. For instance, Park et al. [83] fabricated carbon foam/N-doped graphene@MoS<sub>2</sub> hybrid nanostructures as HER electrodes by using melamine sponge as the raw material to prepare carbon foam and N-doped graphene@MoS<sub>2</sub> as the active component (Fig. 9a), and the as-prepared electrode achieved a low  $\eta_{10}$  of  $-170$  mV in 0.5 M H<sub>2</sub>SO<sub>4</sub>. Another unique advantage of carbon foam is that heteroatom





**Fig. 8.** (a) Photograph (left) and SEM (right) images of graphene aerogel decorated with MoS<sub>2</sub>. Panel (a) reproduced with permission from Ref. [30] © Royal Society of Chemistry. (b) Preparation procedure of GA-MoS<sub>2</sub> composite. Panel (b) reproduced with permission from Ref. [79] © Wiley. (c) Fabrication process for the graphene aerogel electrode decorated with cobalt phosphide nanoparticles. Panel (c) reproduced with permission from Ref. [21] © Wiley. (d) SEM image and photograph of carbon fiber aerogel. Panel (d) reproduced with permission from Ref. [80] © American Chemical Society.

doping can be concurrently achieved with the formation of a 3D foam network during pyrolysis. For example, Wang et al. [84] prepared a Ni<sub>3</sub>C nanoparticles-embedded carbon foam electrode (Ni<sub>3</sub>C@PCN) by carbonizing the mixture of citric acid monohydrate and nickel acetate tetrahydrate, leading to the formation of a foam network and the incorporation of Ni<sub>3</sub>C nanoparticle. Significantly, the as-prepared Ni<sub>3</sub>C@PCN electrode exhibited an E<sub>onset</sub> of -65 mV and η<sub>20</sub> of -203 mV in acidic media (Fig. 9b–e), and also showed only a negligible loss of the current density during 1000 cycles.

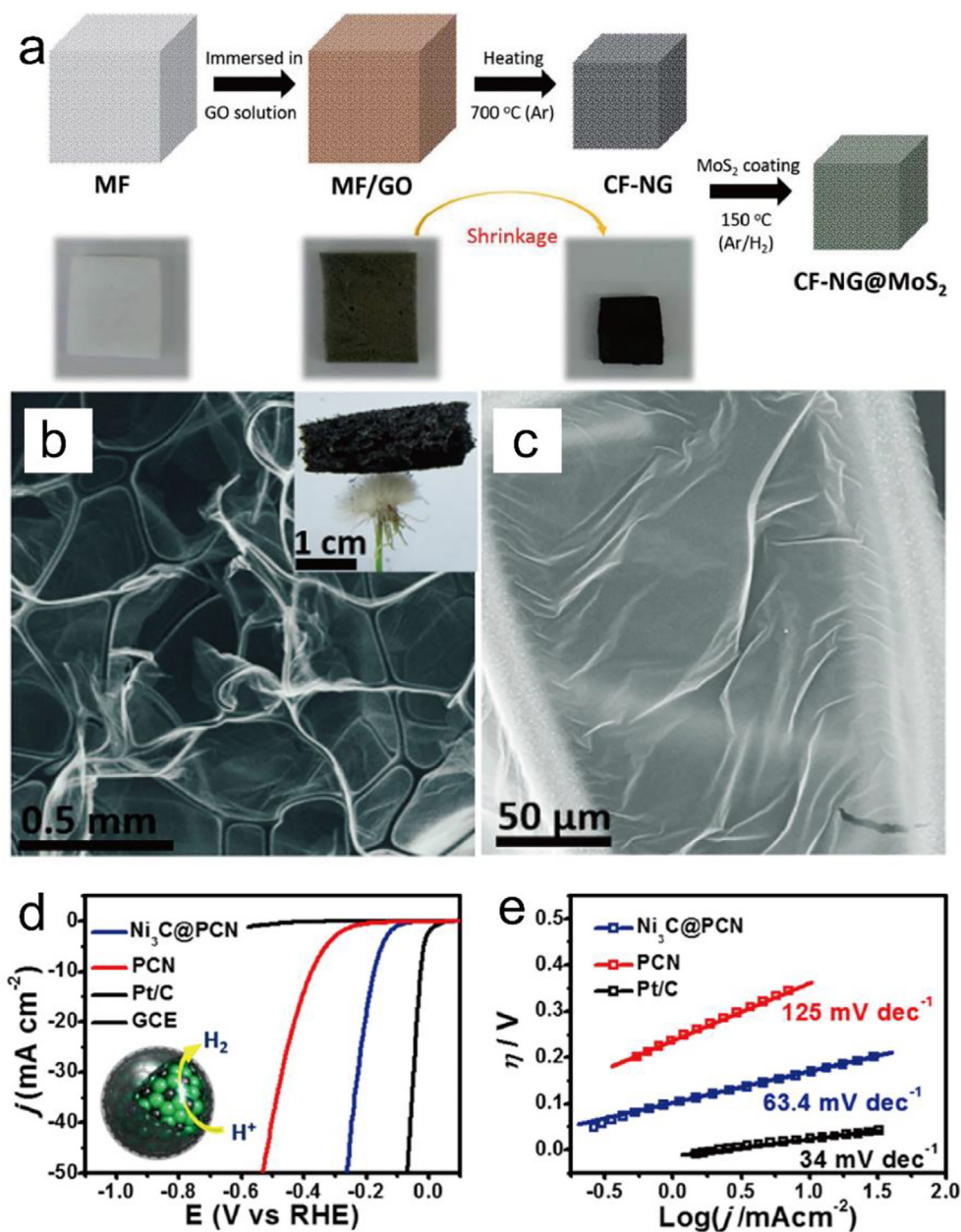
The carbon foam electrode prepared by pyrolysis is usually much more stable than that formed through π-π interaction between carbon precursors, providing a stronger mechanical performance. Unlike a multistep synthesis process, direct pyrolysis does not require rigorous conditions or high-cost templates. However, one challenge remains: the formation of a porous structure in carbon foam mainly relies on the raw material and the pyrolysis process; and the lack of control in pyrolysis can result in a low degree of porosity and hence hinder the penetration of electrolyte in the interior space of carbon foam electrode.

#### 4. Other emerging electrodes

As discussed above, porous electrodes with a large surface area, abundant mesoporosity/macroporosity and a conductive skeleton can provide multiple pathways to facilitate fast ion diffusion and electron transfer. However, the low availability and controllability of pore structure in the electrode are two crucial limitations which can not be resolved yet by chemical methods, such as pyrolysis, hydrothermal treatment, and templates. Interestingly, 3D printing technology has emerged as an effective fabrication tool for rapid prototyping and can achieve a rational electrode design for an accurate control of electrode

porosity. This can effectively increase the formation of pores inside the electrode and facilitate ion and bubble transport. Generally, computer numerical control additive manufacturing of 3D printing can take advantage of various precursor materials, such as polymer, graphene oxide and stainless steel, allowing for the fabrication of products with different physical, chemical, mechanical, and electrical properties [87–89]. For example, Ambrosi et al. [90] prepared stainless-steel electrodes using selective laser melting additive manufacturing and then electrochemically modified the electrode surface to alter the surface composition and tune its catalytic properties by depositing Pt as the HER catalyst and IrO<sub>2</sub> as the oxygen evolution reaction (OER) catalyst. Experimentally, the as-prepared electrode was then used in an electrolyzer in 1 M KOH, which achieved efficient hydrogen and oxygen evolution with a small voltage of 1.5 V (Fig. 10). This study strongly demonstrated the feasibility of building electrodes and electrochemical cells via 3D metal and polymer printing [90]. Nevertheless, despite many advantages, 3D printed electrodes based on carbon materials have not been reported so far. Further research is strongly desired.

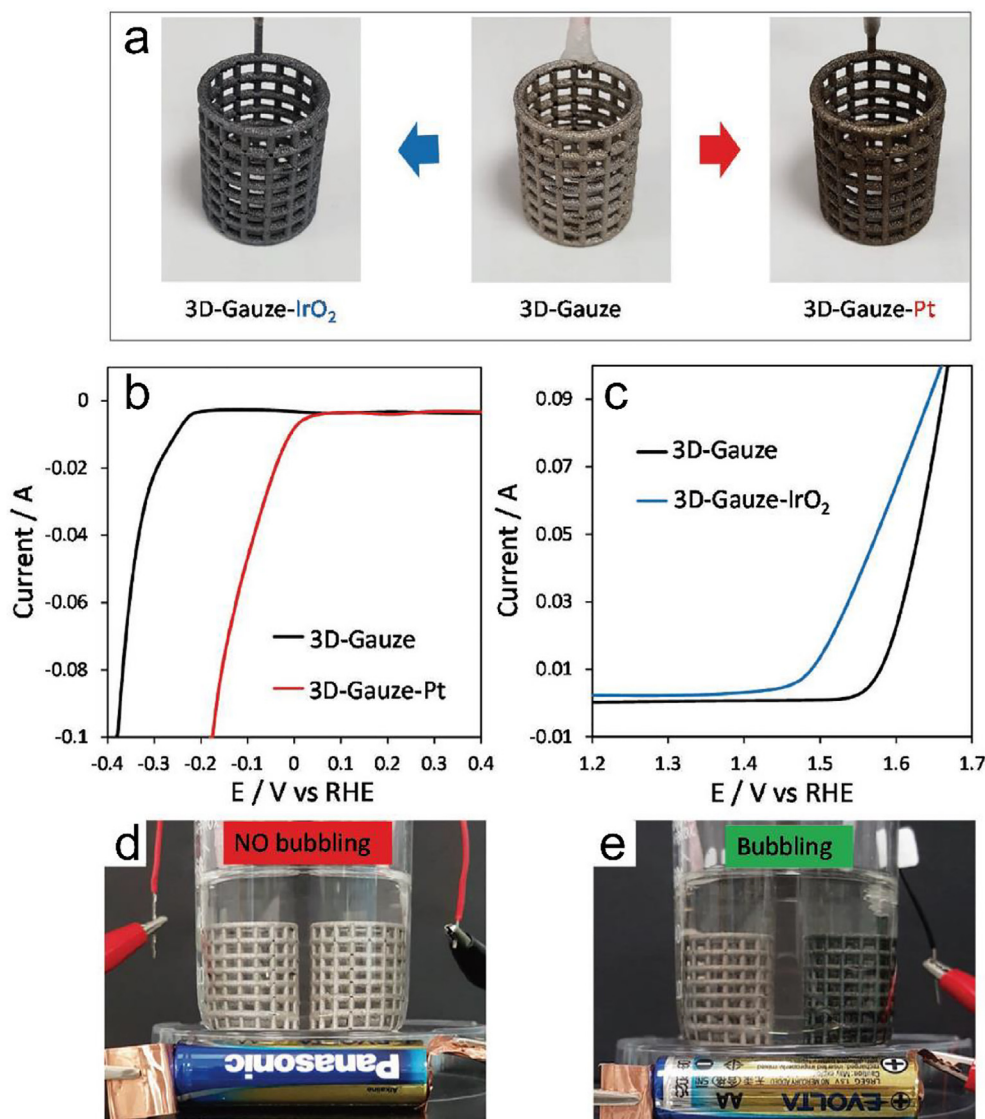
Furthermore, current studies mainly focus on the improvement of the electroactive surface area, electron transfer and ion diffusion in the HER electrode through the optimization of pore structure, heteroatom doping and nanoparticles incorporation [83,91–95]. However, as HER is an electrochemical process involving also ion diffusion and bubble formation, transport and release, the eventual HER performance is actually determined by these factors combined. Improved ion diffusion and electron transfer will effectively facilitate the hydrogen evolution rate and the formation of hydrogen bubbles on the electrode surface. However, the massive bubbles adhered on the surface of electrode will decrease the effective surface area and hinder the electron-transfer reaction and ion diffusion, thus conversely deteriorating the HER



**Fig. 9.** (a) Preparation procedure for carbon foam/N-doped graphene@MoS<sub>2</sub> hybrid. Panel (a) reproduced with permission from Ref. [83] © Royal Society of Chemistry. (b, c) SEM images of Ni<sub>3</sub>C@PCN (inset to panel b is a photograph of the sample), (d) polarization and (e) Tafel curves of varied electrodes. Panels (b-e) reproduced with permission from Ref. [84] © American Chemical Society.

performance. Therefore, the release of hydrogen bubbles from the electrode surface is a key point to improve both effective electroactive surface area and electron transfer and ion diffusion. The bubbles on the electrode surface experience three forces: buoyance force which is proportional to the bubble volume, gravity which is proportional to the mass of the hydrogen bubbles, and adhesion force from the underlying catalyst film. Among these, gravity is negligible due to the much lower density of hydrogen bubbles compared to the electrolyte. It is generally accepted that the detachment of bubbles requires a buoyance force higher than adhesion forces. Therefore, a higher adhesion force would cause a larger detachment diameter and adhesion surface area of bubbles on the electrode. “Superaerophobic” surfaces have been reported as an effective structure to decrease the adhesion force and to diminish the negative effects caused by the adhered hydrogen bubbles. It has been reported that the adhesion force originates from solid-liquid-gas TPCL, and nanostructured surface can create a discontinuous

TPCL and hence decrease the adhesion force of bubbles on the electrode surface, leading to a small release size and fast removal of hydrogen bubbles at high reaction rates (Fig. 11a-c). For example, Lu et al. [28] constructed a MoS<sub>2</sub> nanostructured HER electrode with a gas bubble contact angle of  $153.6 \pm 2.4^\circ$ , which is higher than  $135.2 \pm 3.3^\circ$  for a flat MoS<sub>2</sub> film electrode. Adhesive force measurements with 3 μL bubbles revealed that the as-prepared electrode exhibited a low adhesive force of  $10.8 \pm 1.7 \mu\text{N}$ , as compared to  $124.8 \pm 6.1 \mu\text{N}$  for flat MoS<sub>2</sub> film. Eventually, H<sub>2</sub> bubbles on the MoS<sub>2</sub> nanostructured electrode were detached with an average diameter less than 100 μm (> 500 μm for flat MoS<sub>2</sub> film) (Fig. 11d and e). Therefore, surface nanoengineering is an effective approach to reduce the solid-gas interaction in electrode fabrication, and to facilitate the release of hydrogen bubbles and ion diffusion. Indeed, nanostructures, such as nanowires, nanorods, nanoflakes, nanosheets, nanoflowers and nanotube arrays, have been widely used for the fabrication of superaerophobic



**Fig. 10.** (a) Photographs of 3D-printed gauze steel electrodes with electrochemical surface modification, (b) HER and (c) OER polarization curves in 1 M KOH, video snapshots of water splitting operations in 1 M KOH using (d) bare 3D-gauze electrodes as the anode and cathode and (e) Pt-modified 3D-gauze as the cathode and IrO<sub>2</sub>-modified 3D-gauze as the anode. Reproduced with permission from Ref. [90] © Wiley.

HER electrodes [28,29,96–98]. The mechanism of superaerophobic surface construction can also be expanded to other gas evolution reactions, such as oxygen evolution, and chlorine evolution [99,100].

Apart from the release of hydrogen bubbles from electrode surface, the transport of bubbles in the porous structure of electrode has received negligible attention thus far. Actually, as the charge transfer kinetics on the catalyst surface have been substantially improved, the HER overpotential may be dominated by mass transport, especially at a high current density. The transport of bubbles inside the porous structure of the electrode, such as nickel/copper foam, not only affects the effective surface area in the interior of electrode, but also influences the penetration of electrolyte. Reasonably, it can be understood that the bubbles may be adhered or plugged in the pores of electrode when the bubble size is larger than the pore size. In contrast, the bubbles are likely to diffuse out of the electrode when the bubble size is smaller than the pore size of the electrode. For example, Xu et al. [32] synthesized nanostructured NiMo alloy on metal foam as a superaerophobic electrode, and observed that the hydrogen bubbles were generated and left quickly with a small average releasing diameter of ca. 32  $\mu\text{m}$ . However, in the control electrode, the as-produced H<sub>2</sub> bubbles grew gradually in size and plugged severely within the skeleton

of the metal foam electrode. In fact, the as-obtained electrode delivered an excellent performance that was about three times better than those of commercial Pt/C and IrO<sub>2</sub>/C catalysts-based electrodes at 1.9 V. Therefore, the pore size and bubble release diameter can be critical factors determining the bubble removal in the interior of the electrode. Inevitably, superaerophobic pores will decrease the release size and facilitate the removal of bubbles from the pores. Thus, rational design of the pore structure is a feasible method to prevent the blocking of bubbles in the electrodes in the practical applications.

## 5. Conclusions and perspectives

In this review, we analyze the correlation between electron transfer, ion diffusion, bubble release within the electrode and the corresponding electrochemical HER performance. With an apparent intrinsic activity of the catalysts, ion diffusion and bubble release stand out as two critical factors that may dictate the eventual HER performance. Thus, in this review we summarize recent progress in the development of effective strategies for the design and fabrication of HER electrodes, and compare the HER performance in these fabrication methods within the context of ion diffusion, bubble release, and HER performance. Such

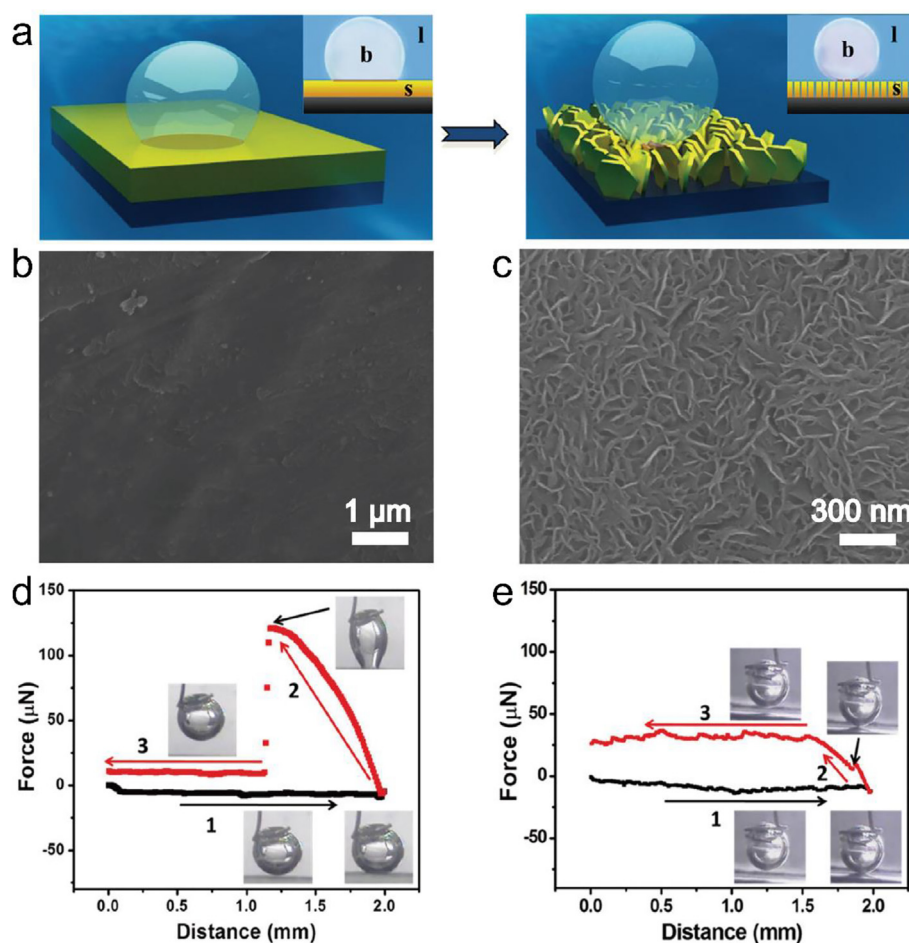


Fig. 11. (a) Schematic illustration of adhesion behaviors of gas bubbles on flat film (left) and nanostructured film (right). Insets are the side views to show the different intact and discontinuous TPCL on flat and nanostructured films, and the different contact angles. (b) and (c) SEM images of flat and nanostructured MoS<sub>2</sub>, (d) and (e) adhesive forces measurements of the gas bubbles on flat and nanostructured MoS<sub>2</sub> films. Reproduced with permission from Ref. [28] © Wiley.

fundamental insights are critical in the utilization of the catalysts for the construction of high-performance HER electrodes.

Nevertheless, despite apparent progress, challenging issues remain. For instance, in practical applications, the consumption of protons will not only lead to alkalinization of the electrolyte on the cathode surface but also pH polarization between the anode and cathode. This necessitates the development of high-efficiency HER electrodes in a wide pH range that can overcome the increasing concentration overpotential. In addition, further research is needed to advance the technology of 3D printing for the preparation of carbonaceous HER electrodes so as to take advantage of the extensive progress of carbon-based HER catalysts. Furthermore, the durability and sustainability of superaerophobic surfaces is another critical issue of HER electrodes, because vibration, stress and electrolyte flow may destroy the electrode structure during long-term operation. Further breakthroughs are urgently needed in the design, construction and nanoengineering of the electrodes.

#### Author contributions

The manuscript was written through contributions of all authors. All authors have given approval to the final version of the manuscript.

#### Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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